Soluble and Amorphous, Phenyl-Substituted Poly(1,4-xylylene) by Chemical Vapor Deposition

O. Schäfer and A. Greiner*

Philipps-Universität Marburg, Fb Physikalische Chemie/Polymere und Wissenschaftliches Zentrum für Materialwissenschaften, Hans-Meerwein-Strasse, D-35032 Marburg, Germany

Received May 9, 1996 Revised Manuscript Received June 17, 1996

Introduction. Poly(1,4-xylylene) (PPX) and its derivatives have gained significant interest for fundamental research as well as for technical applications. 1-3 Vapor phase pyrolysis at reduced pressure of paracyclophanes⁴ or diesters of α,α' -dihydroxy-1,4-xylenes⁵ or α,α' -dibromo-1,4-xylenes in the presence of Zn/Cu⁶ yields monomers, predominantly 1,4-quinodimethanes, which can form PPXs upon chemical vapor deposition (CVD) on cold substrates. This process is generally called the vapor coating process. The products of this process are solvent- and pinhole-free PPX films. Many of these PPXs exhibit very good solvent resistance, high melting points, and high degrees of crystallinity. Although soluble derivatives of poly(1,4-xylylene) were prepared by the vapor coating process, molecular weight data are based upon counting of radical chain ends by ESR spectroscopy or on viscosimetry. To the best of our knowledge, no molecular weight data are published for PPXs prepared via CVD based on gel permeation chromatography (GPC) or light scattering. The knowledge of the molecular weights and their distributions obtained by CVD is of interest for understanding the mechanism of the chain propagation as well as for applications which rely on high molecular weight material. We basically want to understand how regioselectivity and molecular weight distribution can be controlled in the polyreactions proceeding via 1,4-quinodimethanes.

The concept of improving the solubility of aromatic polymers as well lowering their crystallinity by lateral phenyl substituents has been shown for aromatic polyesters, polyaramides, and poly(phenylenevinylenes). Phenyl-substituted PPX (PPPX) seems to be an ideal candidate for obtaining a readily soluble amorphous PPX combined with very good thermal resistance predominantly limited by the sensitivity of the methylene moieties. The PPPX obtained by the vapor coating process with 2-phenyl-1,4-xylylene was insoluble and intractable, 10 which is most likely due to a nonlinear structure. Therefore we tried to synthesize PPPX applying phenyl-substitued diesters of α,α' -dihydroxy-1,4-xylenes but were not successfull.¹¹ Based on our previous experiments with α , α' -dihalogeno-1,4-xylenes, where we obtained films of α -halogeno-substituted PPXs as precursors for PPV by the vapor coating process, 12 we report here on a new synthetic approach to PPX and PPPX by CVD using α -bromo-1,4-xylene as starting

Experimental Section. Synthesis of α-**Bromo-2-phenyl-1,4-xylene (2).** A 500 mL flask was charged with 50 g (275 mmol) of 2-phenyl-1,4-xylene (1)¹³ and 300 mL CCl₄. The mixture was heated to reflux and 12.8 mL (250 mmol) of bromine was added dropwise while the reaction mixture was illuminated with a 500 W lamp. Heating was continued for 30 min after

addition of bromine and then cooled to room temperature. The mixture was subsequently washed with aqueous NaSO₃, aqueous NaCO₃, aqueous NaCl, and H_2O and then dried over MgSO₄. After removal of CCl₄, the crude product was fractionated twice by vacuum distillation, bp 105 °C, 10^{-3} Torr. Yield of **2**: 51 g (197 mmol), 72% (based on **1**).

IR (KBr, cm⁻¹): 3059, 3026, 2962, 2973, 2921, 2859, 1952, 1903, 1812, 1766, 1610, 1569, 1488, 1443, 1379, 1270, 1220, 1147, 1113, 1074, 1024, 885, 822, 779, 745, 730, 701.

¹H-NMR (CDCl₃, δ/ppm): 2.28 (s, 0.6 H), 2.40 (s, 2.4 H), 4.46 (s, 1.6 H), 4.52 (s, 0.4 H), 7.10–7.50 (m, 8 H). Anal. Calcd for C₁₄H₁₃Br (M= 261.16 g/mol): C, 64.39; H, 5.02. Found: C, 64.26; H, 5.13.

Synthesis of Poly(phenyl-1,4-xylylene) (3) by CVD. The quartz boat in the vaporization zone of the preheated (900 °C) pyrolysis apparatus (see ref 12) was charged under an argon atmosphere with 1.89 g of 2 (7.2 mmol). The apparatus was evacuated, and the pressure was kept between 0.2 and 0.4 mbar. Films of 3 were formed at the glass walls of the tube behind the pyrolysis zone above the cooling trap. The product was dissolved in THF, reprecipitated in methanol, and dried under vacuum at 80 °C. Yield: 0.91 g of 3 (5.04 mmol), 70% (after subtraction of nonconverted 2). $T_{\rm g} = 107$ °C, $M_{\rm n} = 8700$, $M_{\rm w} = 55\,000$ (both from GPC with chloroform versus polystyrene standards).

IR (KBr, cm⁻¹): 3044, 3012, 2921, 2857, 1613, 1487, 1451, 824, 757, 702. 1 H-NMR (CDCl₃, δ /ppm): 2.23–2.40 (m, 0.23 H), 2.70 (m, 2 H), 2.91–2.98 (m, 2 H), 6.77–7.35 (m, 8 H). 13 C-NMR: 20.92, 34.58, 35.01, 37.37, 37.53, 126.63, 126.75, 127.20, 127.33, 127.46, 127.96, 129.13, 129.28, 129.40, 129.95, 130.10, 130.25, 130.65, 130.80, 136.21, 136.65, 136.79, 139.18, 139.27, 139.39, 139.49, 141.62, 141.79, 141.84. Anal. Calcd for (C₁₄H₁₂)_n (M = 180.25 g/mol): C, 93.29; H, 6.71. Found: C, 93.17; H, 6.35.

Results and Discussion. The bromination of 1 results in two isomeric products (Scheme 1) which can be identified by ¹H- and ¹³C-NMR spectroscopy. The overall yield of 2 was 72% based on the amount of 1. The 2-isomer is obtained according to ¹H-NMR spectroscopy in 57.6% yield and the 3-isomer in 14.4% yield. The proportion of the isomers should be insignificant for the polyreaction to 3. The vapor phase pyrolysis of 2 resulted in the formation of HBr and 3 as transparent coatings in 70% yield of 3 based on converted 2. 3 obtained via vapor phase pyrolysis of 2 exhibits very good solubility in solvents like THF or chloroform, which is in contrast to the product obtained via vapor phase pyrolysis of 1,4-xylene. 10 The improved solubility allowed structural analysis by ¹H- and ¹³C-NMR spectroscopy in solution and molecular weight analysis by GPC.

The 1 H-NMR spectrum of **3** exhibits large signal groups in the aliphatic region at 2.70–2.98 ppm (Figure 1). Each of the signals can be resolved to at least three individual signals or at least shoulders. The small signals from 2.26 to 2.40 ppm represent presumably methyl end groups. A $P_{\rm n}$ of 26, corresponding to $M_{\rm n}$ = 4700 g/mol, can be calculated from the integrals if each homologue of **3** is terminated by two methyl groups. The signals of the aromatic protons are represented by a multiplet from 6.77 to 7.35 ppm. Focused on one carbon atom of the methylene group four different diad structures are possible (Figure 2), which should result in four signals in the 13 C-NMR spectrum. The 13 C-NMR spec

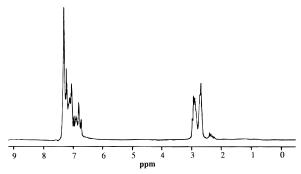


Figure 1. ¹H-NMR spectrum of 3 in CDCl₃.

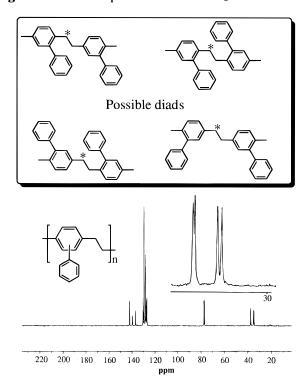


Figure 2. ¹³C-NMR spectrum of 3 in CDCl₃.

Scheme 1 Br₂ / hv 2'-isomer 2 3'-isomer 900°C 0.2-0.4 torr HBr

trum of **3** indeed shows four signals in the aliphatic region between 34.58 and 37.53, which is consistent with a statistical arrangement of the phenyl substitutent along the polymer backbone of **3** (Figure 2). This means that the bulky phenyl substituent does not induce any regioselectivity in the polyreaction of the phenyl-substituted 1,4-quinodimethane. The small signal at 20.92 ppm was attributed to alkyl end groups.

A $M_{\rm w} = 55\,000$ and a $M_{\rm n} = 8700$ was measured for **3** by GPC versus polystyrene standards, which is higher

than found by end group analysis via $^1\mathrm{H-NMR}$ spectroscopy. $M_\mathrm{w}/M_\mathrm{n}$ is monomodal but rather high (6.32), which is obviously due to the reaction conditions. The polyreaction of quinodimethanes in the course of CVD can be regarded as substance polyreaction with high conversions, which is of course limited by the mobility of the reactive chain ends. The corresponding base-induced polyreaction of 2 resulted in a monomodal $M_\mathrm{w}/M_\mathrm{n}=2.37$, which underlines the assumption that the mobility of the reactive chain ends has a significant impact on the width of the molecular weight distribution.

The 5% weight loss of **3** measured by thermogravimetric analysis was detected at 354 °C, which is somewhat lower than for PPX. In contrast to PPX, **3** is amorphous with a $T_{\rm g}$ at 107 °C. PPX exhibits a glass transition at 13 °C¹⁴ and a melting point at 420 °C (with decomposition). The amorphous character of **3** certainly contributed to the improved solubility of **3**.

Conclusions. Phenyl-substituted PPX, which is soluble and amorphous, can be obtained from phenylsubstituted α -bromo-1,4-xylene by vapor phase pyrolysis and subsequent CVD. The enhanced solubility of product 3 allows structural and molecular weight analysis. The phenyl substituents are arranged statistically along the polymer backbone with regard to diad structures. The molecular weight obtained by this new synthetic approach is high enough to produce freestanding films. The liberation of HBr during the vapor phase pyrolysis is certainly a major drawback for coating of sensitive substrates, but the good access to α-halogenated 1,4-xylenes will allow the preparation of a wide variety of PPXs and the investigation of their structure-property relationships, which is presently underway. However, the search for new, less corrosive leaving groups instead of HBr is still a challenge worthy of intensive efforts.

Acknowledgment. The authors are indebted to Prof. Heitz for many helpful discussions and to the Fonds der Chemischen Industrie for financial support.

References and Notes

- Beach, W. F.; Lee, C.; Bassett, D. R.; Austin, T. M.; Olson, R. In *Encyclopedia of Polymer Science and Technology*, New York, 1989; pp 990–1025.
- (2) Iwatsuki, S. Adv. Polym. Sci. 1984, 58, 93.
- (3) Greiner, A. In *The Polymeric Materials Encyclopedia*, Salamone, J. C., Ed.; CRC Press: Boca Raton, FL, in press.
- (4) Gorham, W. F. J. Polym. Sci., Part A-1 1966, 4, 3027.
- (5) Simon, P.; Greiner, A. Polym. J. 1992, 24, 1317.
- (6) You, L.; Yang, G.-R.; Lang, C.-I.; Wu, P.; Moore, J. A.; McDonald, J. F.; Lu, T. M. Mater. Res. Soc. Symp. Proc. 1993, 282, 593.
- (7) Krigbaum, W. R.; Hakemi, H.; Kotek, R. *Macromolecules* 1985, 18, 965.
- (8) Jadhev, J. Y.; Krigbaum, W. R.; Preston, J. *Macromolecules* 1988, 21, 538.
- (9) Greiner, A.; Heitz, W. Makromol. Chem., Rapid, Commun. 1988, 9, 581.
- (10) Auspos, L. A.; Hall, L. A. R.; Hubbard, J. K.; Kirk, W. M.; Schaefgen, J. R.; Speck, S. B. *J. Polym. Sci.* **1955**, *15*, 9.
 (11) Simon, P.; Mang, S.; Schäfer, O.; Greiner, A., manuscript
- (11) Simon, P.; Mang, S.; Schäfer, O.; Greiner, A., manuscript in preparation.
- (12) Schäfer, O.; Pommerehne, J.; Guss, W.; Vestweber, H.; Tak, H. Y.; Bässler, H.; Schmidt, C.; Lüssem, G.; Schartel, B.; Stümpflen, V.; Wendorff, J. H.; Spiegel, S.; Möller, C.; Spiess, H. W.; Greiner, A. Synth. Met., in press.
- (13) Land, H. T.; Hatke, W.; Greiner, A.; Schmidt, H.-W.; Heitz, W. Makromol. Chem. 1990, 191, 2005.
- (14) Kirckpatrick, D. E.; Wunderlich, B. Makromol. Chem. 1985, 186, 2595.
- (15) Niegisch, W. D. J. Appl. Phys. 1966, 37, 4041.

MA960681B